



Development and Experimental Validation of an Advanced Non-Linear, Rate-Dependent Constitutive Model for Polyether Ether Ketone (PEEK)

***David J. Quinn, Ph.D., Jorgen Bergstrom, Ph.D.,
Nagi Elabassi, Ph.D., Sam Chow, Michael Heiss***

Veryst Engineering, Needham, MA

dquinn@veryst.com

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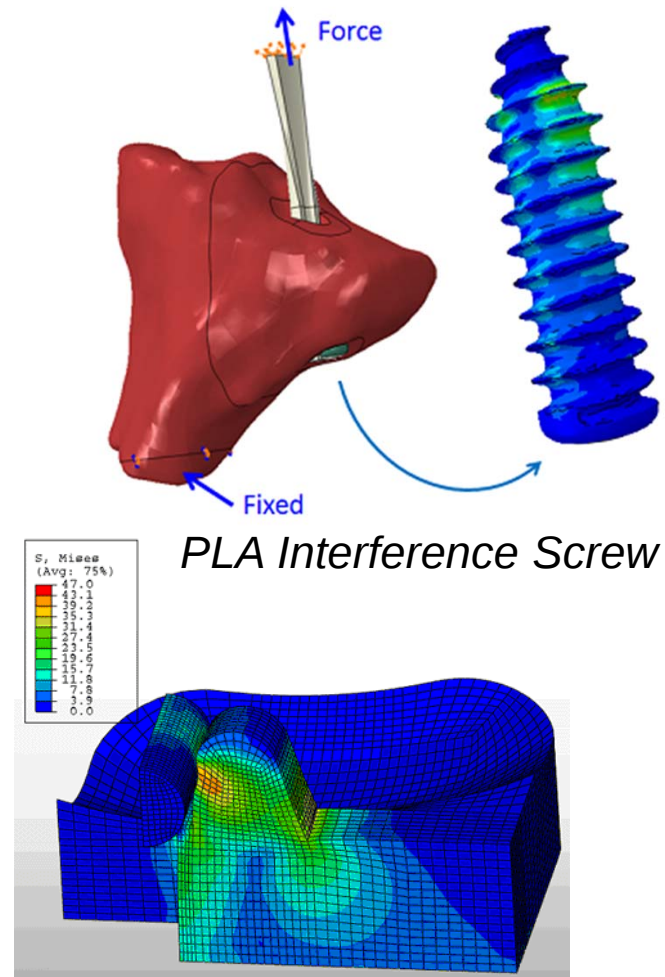
September 2013

Outline

- Motivation: The need for advanced constitutive models of polymers in medical devices.
- Case Study: Calibration and validation of an advanced material model for PEEK.
- Discussion of material model validation.
- Conclusions and future work.

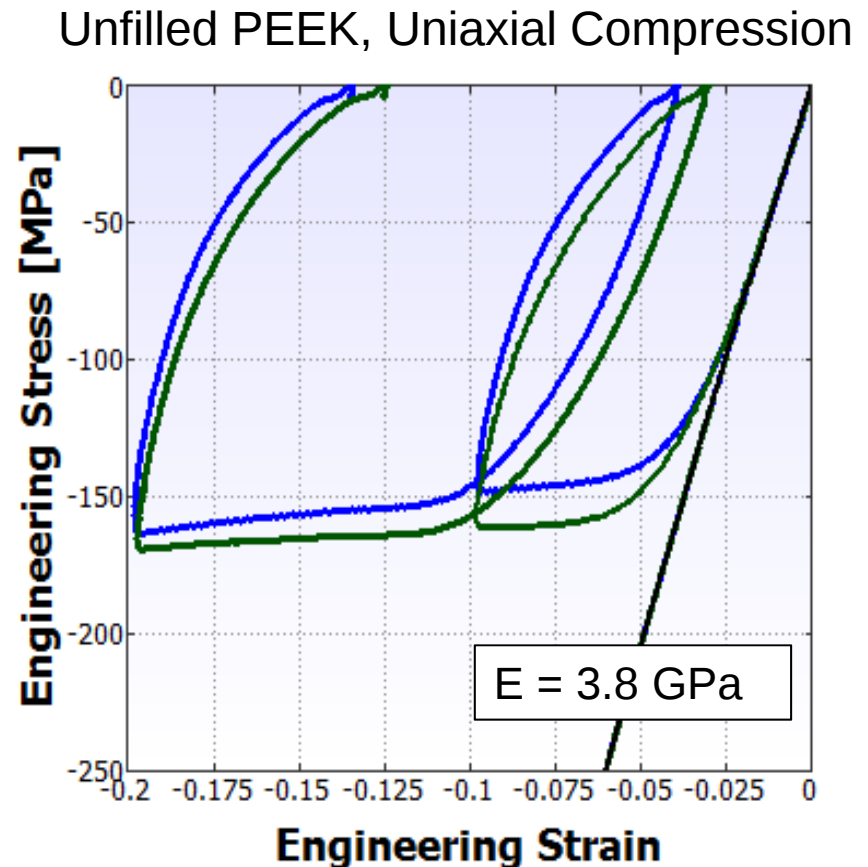
Why do we need an advanced material model?

- Polymers (thermoplastics, rubbers, foams) are not linear, especially above small strains (1-2%)
- Many medical device applications have localized high stresses and strains
- Multiple loading-unloading cycles
- Wide range of time-scales and strain rates



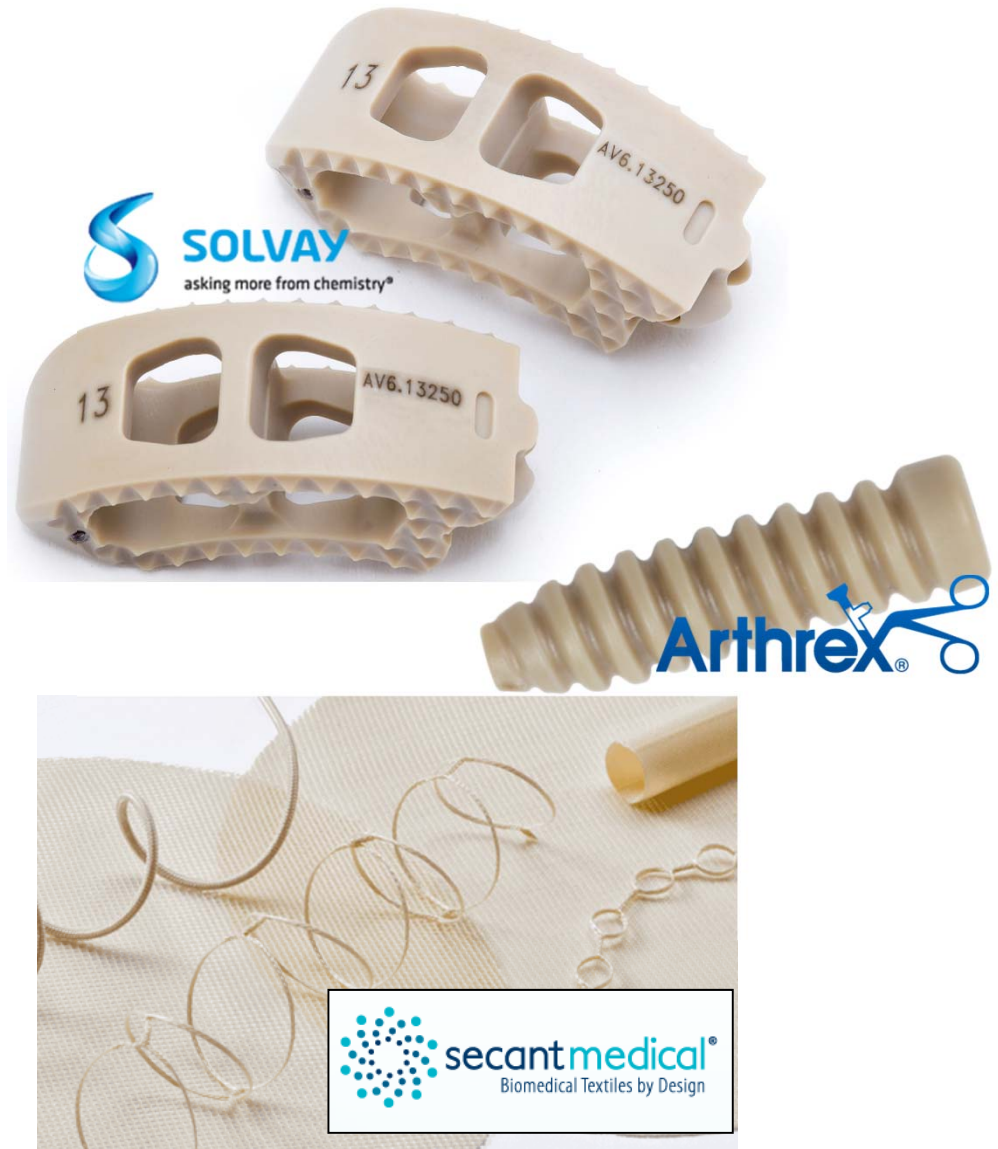
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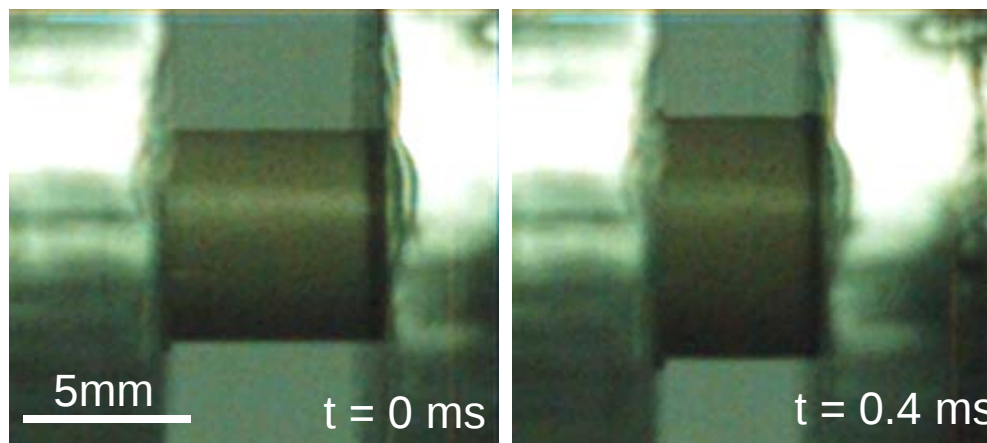
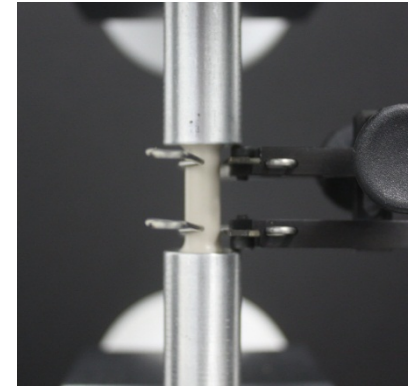
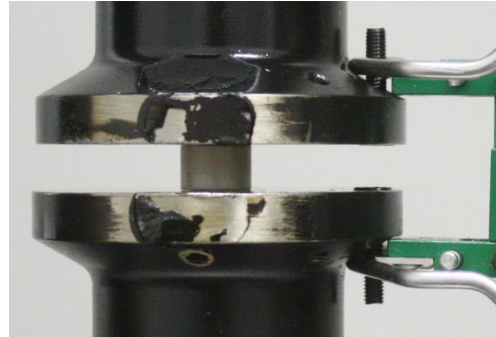
Polyether Ether Ketone (PEEK)

- Good mechanical properties ($E \sim 4\text{GPa}$, $\sigma_{ut} \sim 100\text{ MPa}$)
- Good wear resistance
- Inert, generally biocompatible
- Orthopedic applications
 - Spinal implants/spacers
 - Fixation (screws, plates etc.)
 - Biomedical textiles (wovens, braids)



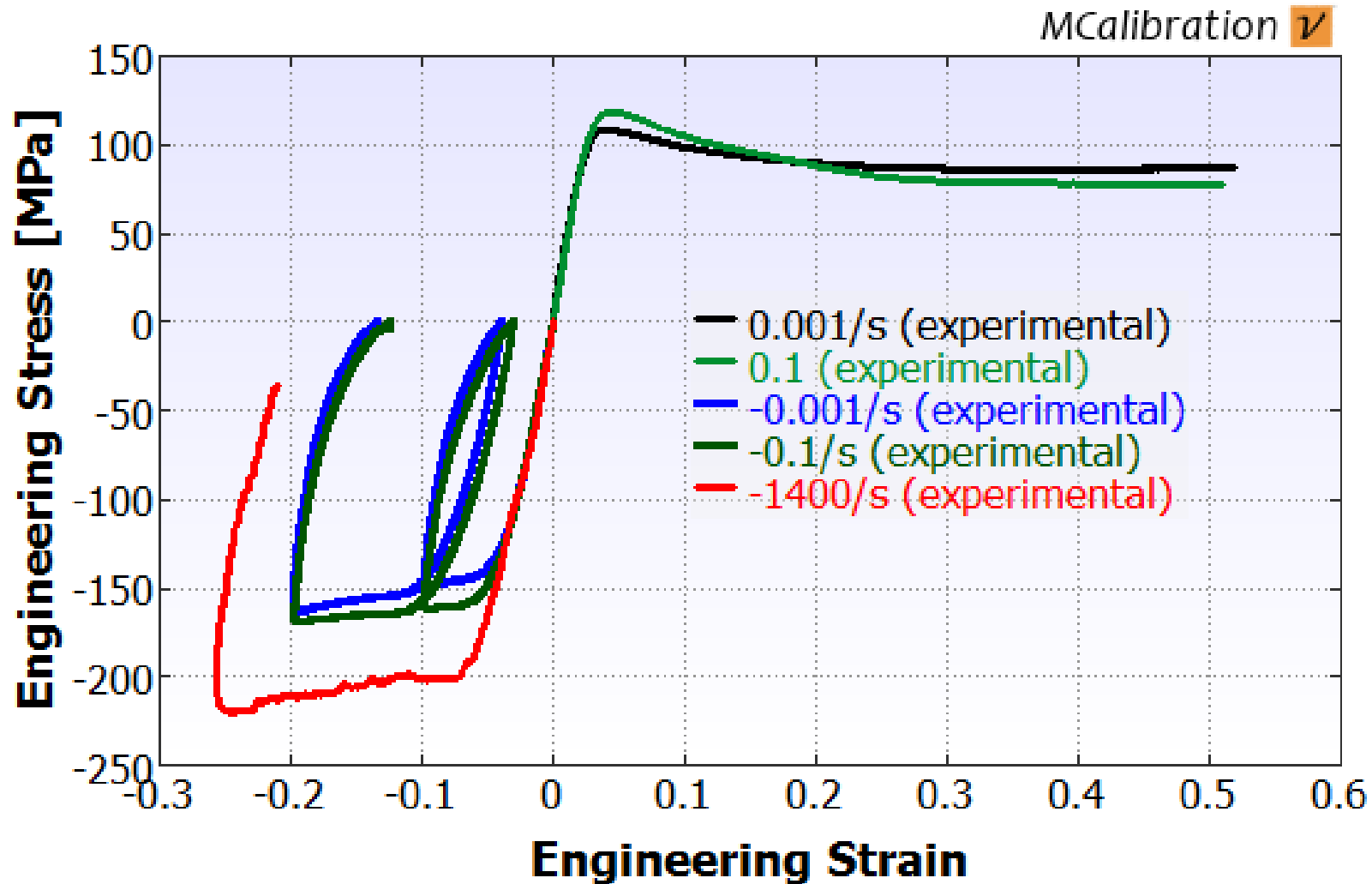
Experimental Testing

- Unfilled PEEK
- Uniaxial compression and tension testing (up to $\sim 0.1/s$)
- Split Hopkinson pressure bar (SHPB) (up to $\sim 1500/s$)



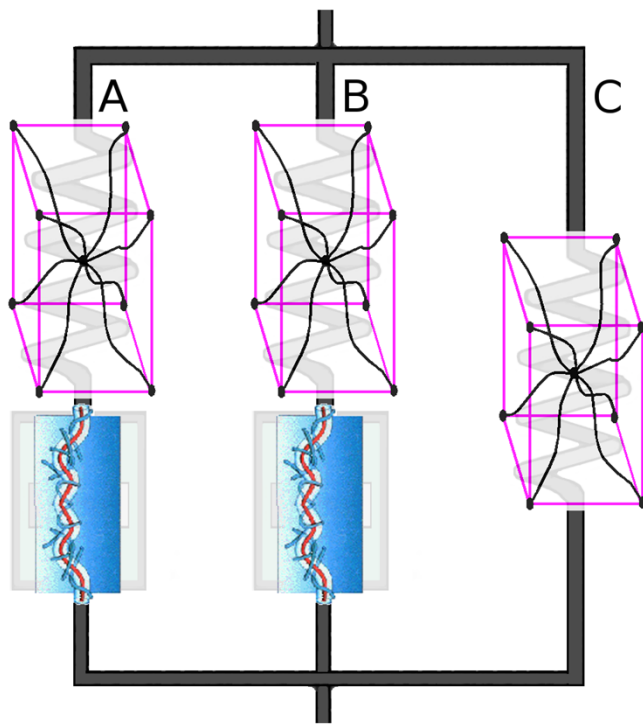
Experimental Results

- Clear rate dependent yield and post-yield behavior

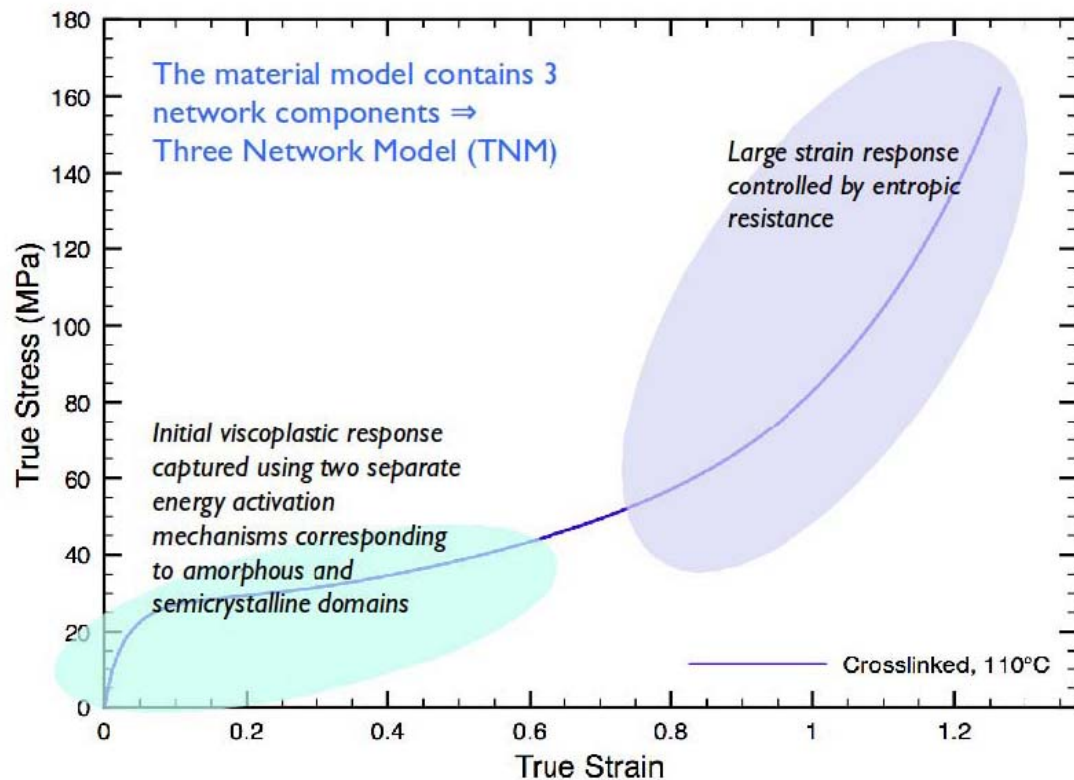


The Three Network Model (TNM)

- Polymer physics-driven modeling framework
- Previously used for modeling UHMWPE
 - Low strain rate over limited range: $\sim 0.001 - 0.01/s$



*Rheological
Representation of the
TNM*



Bergstrom, J.S., Bischoff, J.E., "An Advanced Thermomechanical Constitutive Model for UHMWPE" *Int. J. of Structural. Changes in Solids*, 2(1): 31-39, 2010.

Stress in Networks A & B

Cauchy Stress from Arruda-Boyce 8-chain model (with temp. dependence):

$$\sigma_A = \frac{\mu_A}{J_A^e \overline{\lambda}_A^{e*}} \left[1 + \frac{\theta - \theta_o}{\hat{\theta}} \right] \frac{\mathcal{L}^{-1}(\overline{\lambda}_A^{e*}/\lambda_L)}{\mathcal{L}^{-1}(1/\lambda_L)} \text{dev}[\mathbf{b}_A^{e*}] + \kappa(J_A^e - 1)\mathbf{1}$$

Material Parameters:

μ_A – initial shear modulus
 κ – bulk modulus
 $\hat{\theta}$ – material temp. dependence
 λ_L – locking stretch

Where:

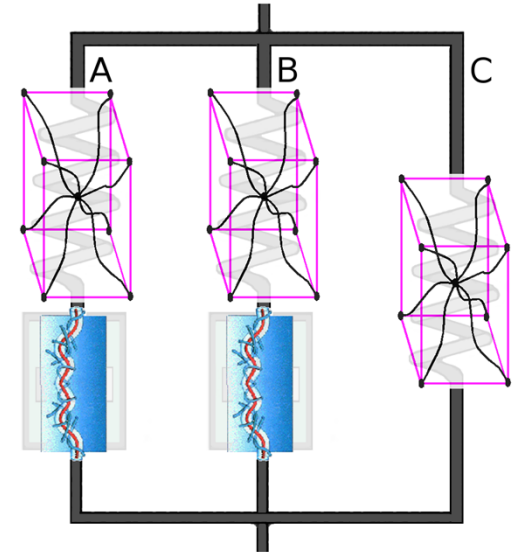
$\mathcal{L}^{-1}$ - inverse Langevin function

θ, θ_o – current, reference temperature

$\overline{\lambda}_A^{e*} = (\text{tr}[\mathbf{b}_A^{e*}]/3)^{1/2}$ - effective chain stretch

$J_A^e = \det[\mathbf{F}_A^e]$

$\mathbf{b}_A^{e*} = (J_A^e)^{-2/3} \mathbf{F}_A^e (\mathbf{F}_A^e)^T$ - Cauchy-green deformation tensor



Controls the initial modulus and the plastic flow behavior

Rate Kinematics – Networks A & B

Shear Modulus Evolution: $\dot{\mu}_A = -\beta[\mu_A - \mu_{Af}] \cdot \dot{\gamma}_A$

Viscoplastic flow rate: $\dot{\gamma}_A = \dot{\gamma}_o \cdot \left(\frac{\tau_A}{\hat{\tau}_A + aR(p_A)} \right)^{m_A} \cdot \left(\frac{\theta}{\theta_o} \right)^n$

$$\tau_A = \|\sigma'_A\|_F \equiv (tr[\sigma'_A \sigma'_A])^{1/2}$$

Viscoelastic velocity gradient: $\dot{\mathbf{F}}^v_A = \dot{\gamma}_A \mathbf{F}^{e-1}_A \frac{\text{dev}[\sigma_A]}{\tau_A} \mathbf{F}$

Material Parameters:

μ_{Af}, μ_{Bf} – final shear moduli
 κ – bulk modulus
 $\hat{\theta}$ – material temp. dependence
 λ_L – locking stretch

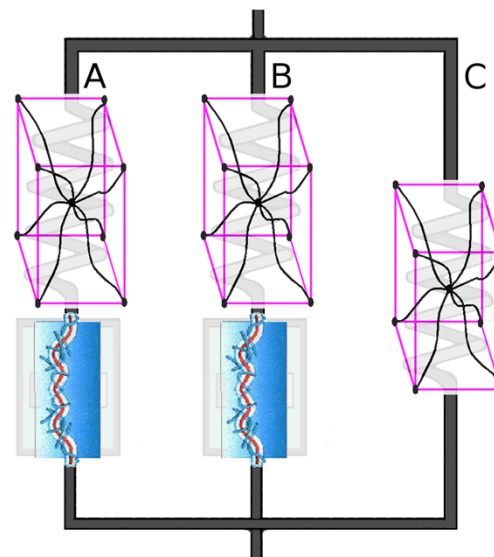
Stress in Network C

Cauchy Stress from eight-chain model with first order I_2 dependence:

$$\sigma_C = \frac{1}{1+q} \left\{ \frac{\mu_c}{J\bar{\lambda}^*} \left[1 + \frac{\theta - \theta_o}{\hat{\theta}} \right] \frac{\mathcal{L}^{-1}(\bar{\lambda}^*/\lambda_L)}{\mathcal{L}^{-1}(1/\lambda_L)} \text{dev}[\mathbf{b}^*] + \kappa(J-1)\mathbf{1} \right. \\ \left. + q \frac{\mu_c}{J} \left[I_1^* \mathbf{b}^* - \frac{2I_2^*}{3} \mathbf{I} - (\mathbf{b}^*)^2 \right] \right\}$$

Controls the large strain response

Total Stress: $\sigma_{Tot} = \sigma_A + \sigma_B + \sigma_C$



TNM – Material model parameters

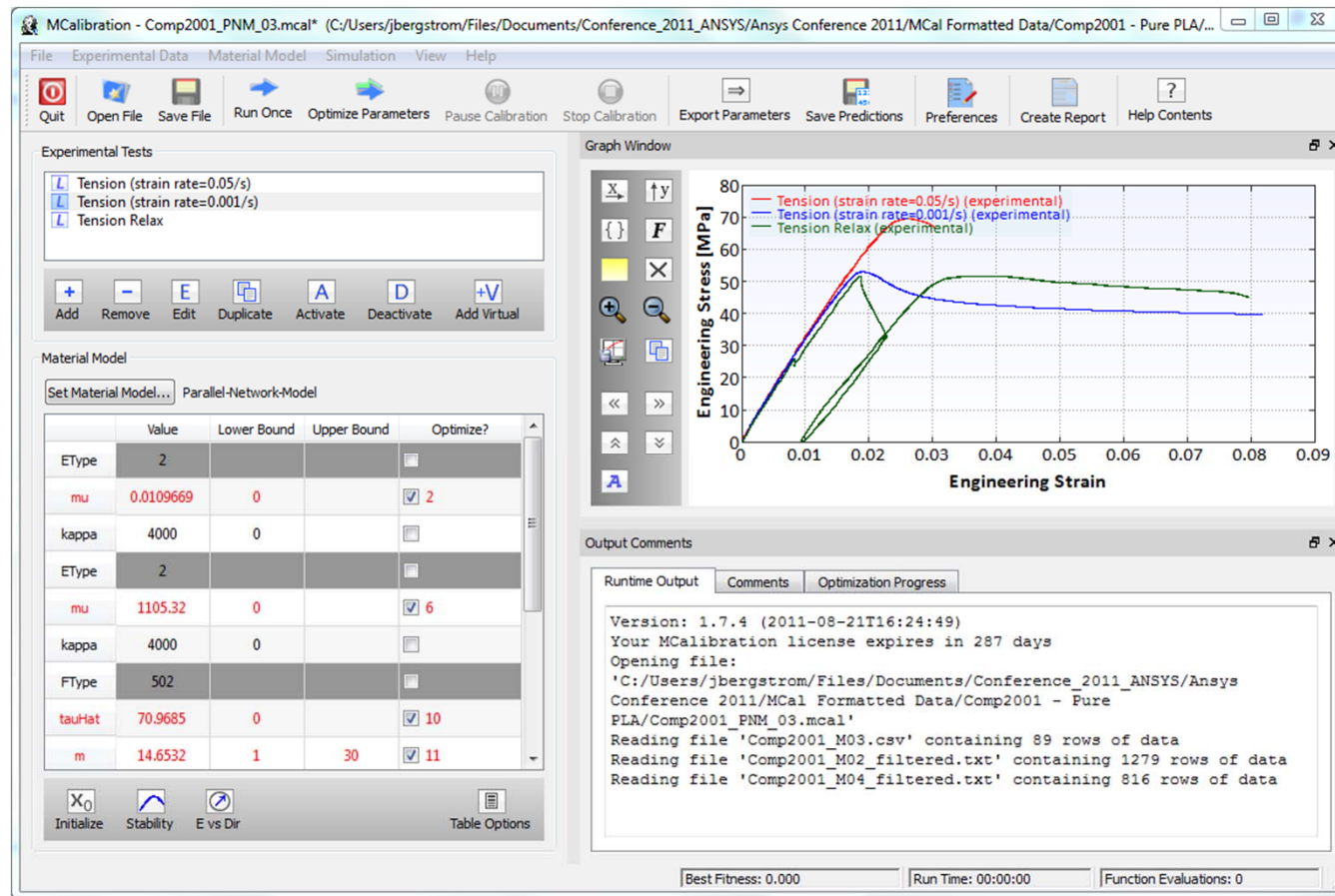
- The TNM is implemented as a user-material model (UMAT) in FEA codes (Abaqus, Ansys)
- Up to 17 material parameters specified or calibrated.
- Calibration requires:
 - Experimental data over range of time-scales
 - Automated process using an optimization method (“Guess and check” will not work)

Table 1: Material parameters used by the three-network model.

Index	Symbol	Umat Name	Unit*	Description
1	μ_A	muA	S	Shear modulus of network A
2	$\hat{\theta}$	thetaHat	T	Temperature factor
3	λ_L	lambdaL	-	Locking stretch
4	κ	kappa	S	Bulk modulus
5	$\hat{\tau}_A$	tauHatA	S	Flow resistance of network A
6	a	a	-	Pressure dependence of flow
7	m_A	mA	-	Stress exponential of network A
8	n	n	-	Temperature exponential
9	μ_{Bi}	muBi	S	Initial shear modulus of network B
10	μ_{Bf}	muBf	S	Final shear modulus of network B
11	β	beta	-	Evolution rate of μ_B
12	$\hat{\tau}_B$	tauHatB	S	Flow resistance of network B
13	m_B	mB	-	Stress exponential of network B
14	μ_C	muC	S	Shear modulus of network C
15	q	q	-	Relative contribution of I_2 of network C
16	α	alpha	T ⁻¹	Thermal expansion coefficient
17	θ_0	theta0	T	Thermal expansion reference temperature

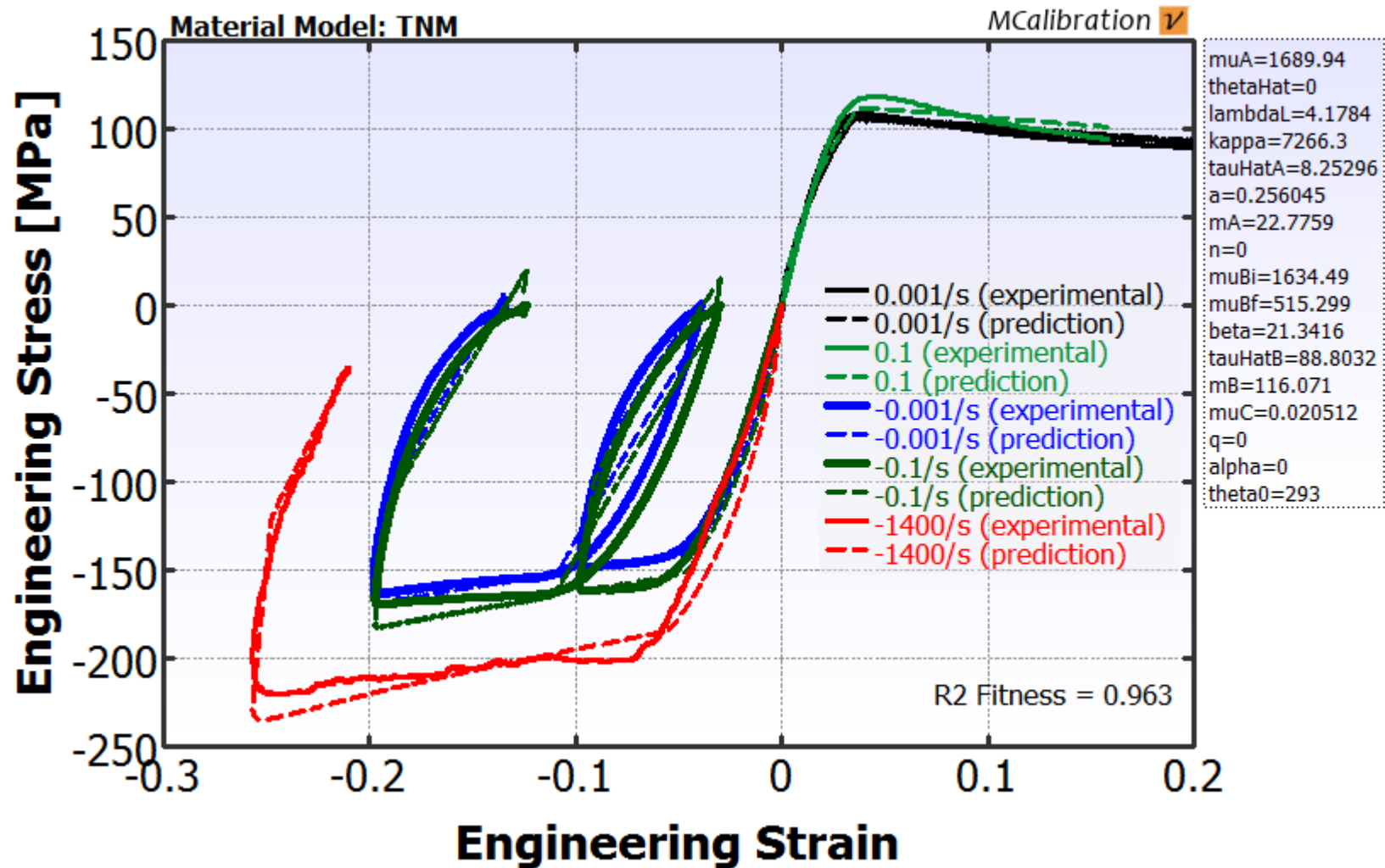
*where: - = dimensionless, S = stress, T = temperature

Material Model Calibration



- The material model calibration was performed using MCalibration.
- MCalibration is commercially available from Veryst Engineering.
- The calibrated material model can be exported for use in Abaqus, Ansys, LS-Dyna with Veryst's PolyUMod Library.

Calibrated Material Model



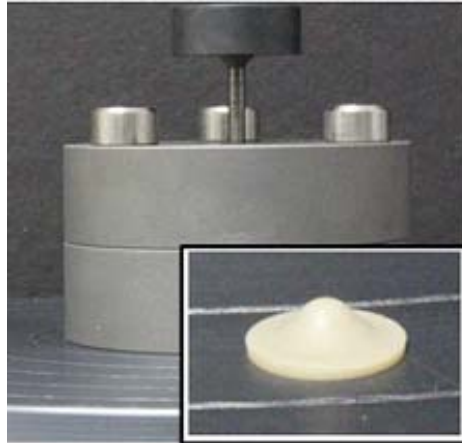
- The calibrated material model shows good agreement with experimental data in tension and compression over 6 decades of strain rate!

The Need for Model Validation Experiments

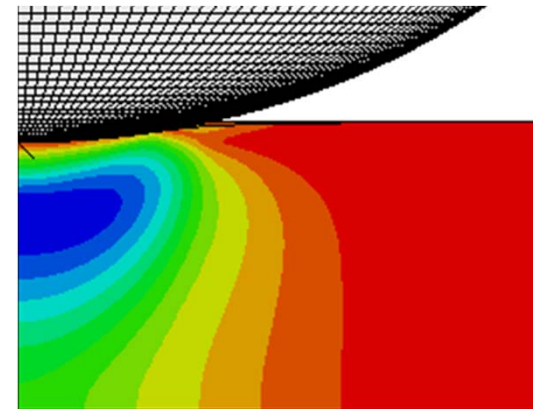
- Constitutive models are typically calibrated using uniaxial tension/compression.
- However, in most applications the material is subjected to multiaxial loading (tension, compression, biaxial, shear).
- Independent experiments should be used to assess the predictive capability of the model in the relevant loading conditions.

Model Validation Experiments

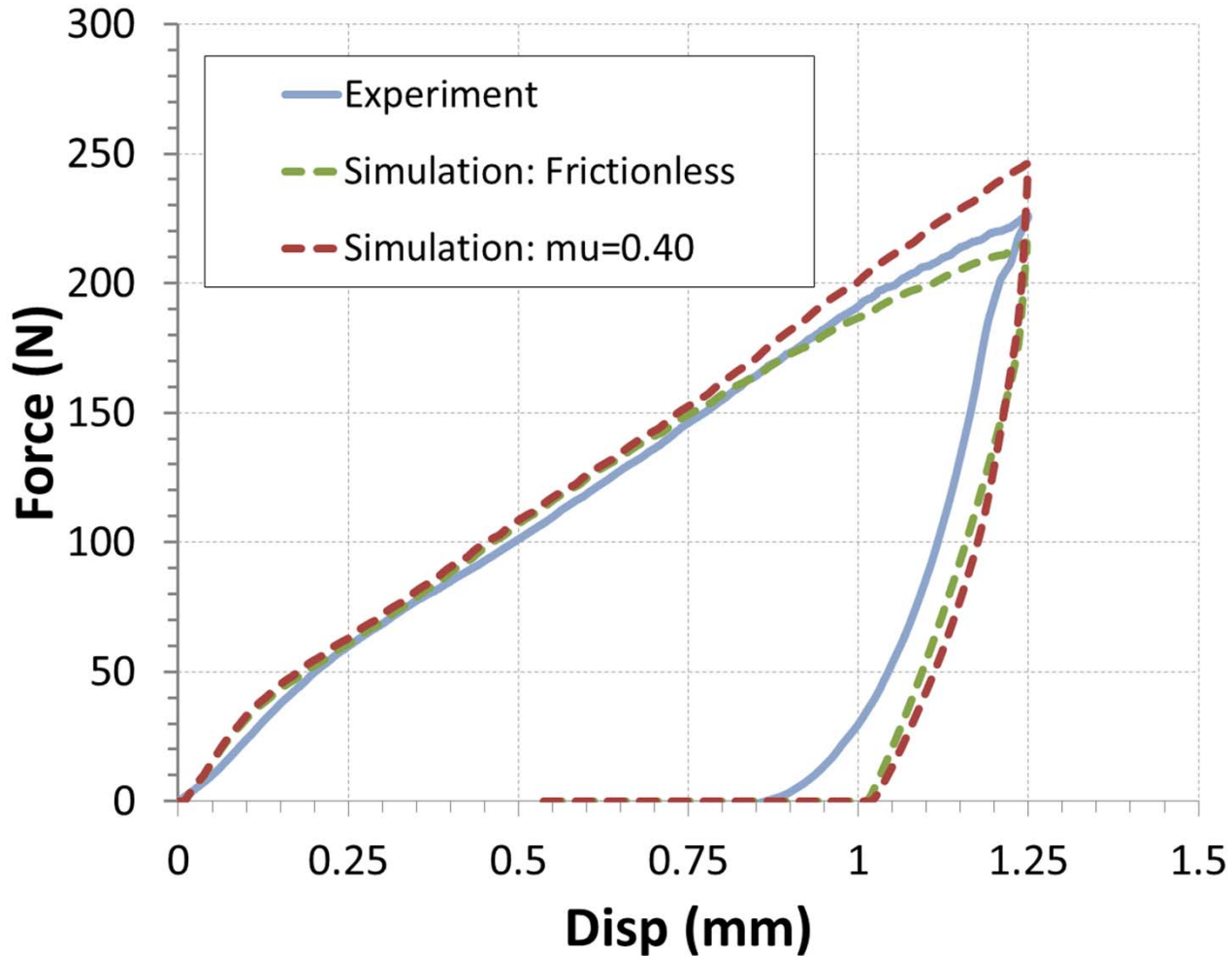
*Small Punch Test
(ASTM F2638)*



*Spherical
Indentation*

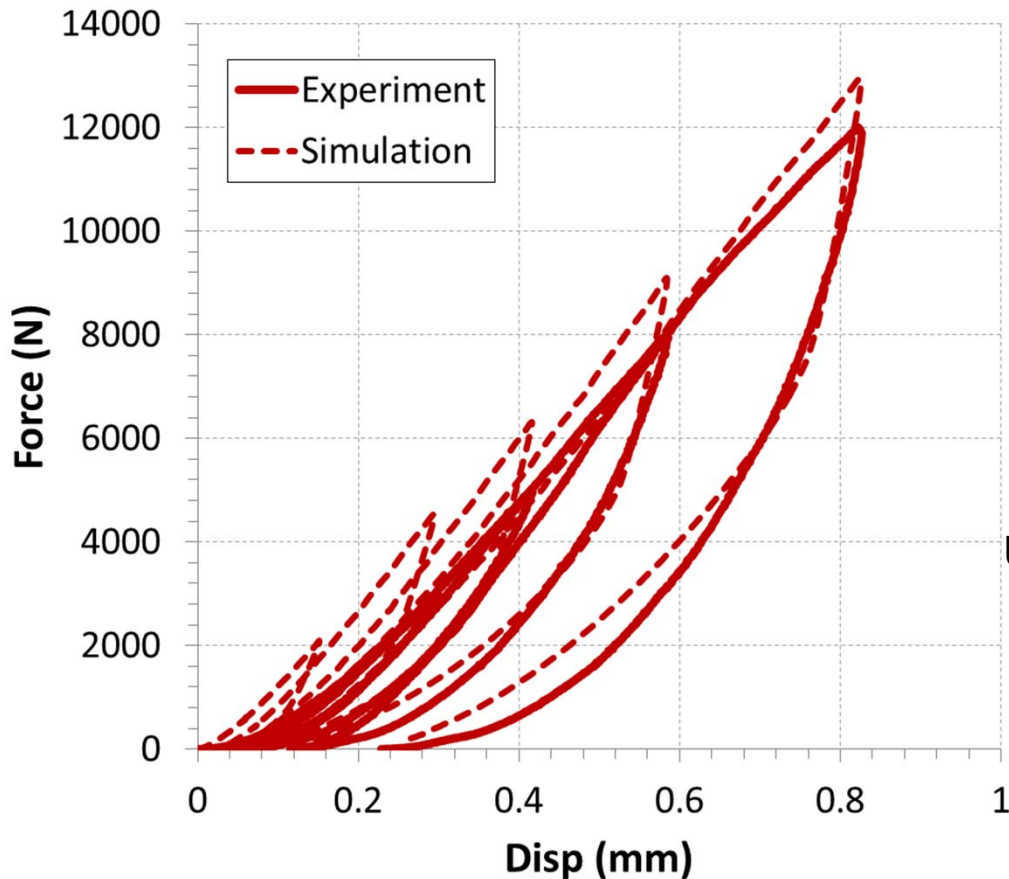


Small Punch Test Results



- Accurate predictions of force, displacement and permanent deformation.
- Results are sensitive to friction at higher loads.

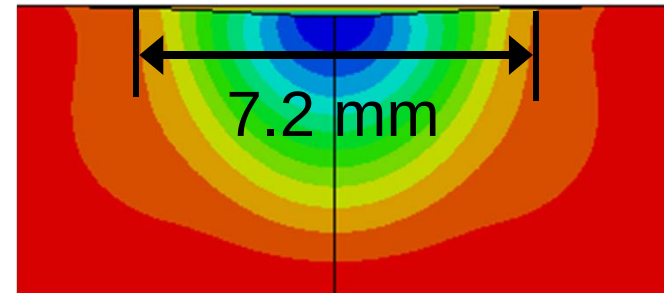
Spherical Indentation Test Results



Experiment, Top View



Unloaded Displacement Field (Side View)



- Accurate predictions of peak force, displacement and deformation.
- Simulation is in-sensitive to frictional effects over these conditions.

When is the Material Model Valid?

Device or Application Specific Questions to Ask:

- What is the dominant stress state and range of stresses/strains in the device/application?
- How do the material model predictions compare over that range ***and slightly beyond?***
- How closely does the validation experiment mimic the anticipated loading environment?
- What is the validation criteria? Total deformation/displacement? “Failure” load?
- How do uncertainties in the material model propagate to the simulation of the device?
- What is the risk (to the patient) of being wrong?

Conclusions

- Accurate simulations of polymeric systems in medical device applications frequently require an advanced, rate-dependent material model.
- When calibrated using an advanced optimization routine, the three network model (TNM) will give accurate stress-strain predictions for PEEK over a large range of strain rates.
- The calibrated TNM can be experimentally validated using independent, multiaxial loading experiments.
- Validation criteria is application and device-specific.

Future Work

- Device-specific application – orthopedics (spinal spacers, bearing surfaces in hip/knee, biomedical textiles).
- Validate model over longer time-scales and strain rates.
- Include damage and failure mechanisms in material model for failure and wear predictions.
- Sensitivity studies and uncertainty predictions.